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Newly synthesised gadolinium bisphthalocyanine sandwich complex: ambipolar organic semiconductor

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Abstract

Time of flight photocurrent transient studies on 5 μ m thick solution processed films of novel non-peripherally octa-octyl-substituted liquid crystalline gadolinium bis-1,4,8,11,15,18,22,25-octakis(octyl) phthalocyanines (8GdPc₂) provide a quantitative analysis of the intrinsic ambipolar charge transport relative to mesomorphic structure of this lanthanide compound. Characteristic liquid crystalline phases of these molecules have been identified from differential scanning calorimetry supported by observation from the UV–visible absorption, showing crystal-columnar mesophase and columnar mesophase-isotropic liquid transitions at 64.2 °C and 162 °C, respectively. The TOF carrier mobility is found to be structure dependent and highest values of $4.73 \times 10^{-6} \,\mathrm{m^2\,V^{-1}\,s^{-1}}$ and $1.6 \times 10^{-6} \,\mathrm{m^2\,V^{-1}\,s^{-1}}$ have been estimated for hole and electron mobilities for hexagonally packed, columnar structures of the spin-coated films. These results are exploitable for development of single molecule based all organic complimentary analogue and digital circuits with tunable field effect performance.

Keywords: discotic liquid crystalline, time of flight, drift mobility, field lowering coefficient, discotic liquid crystal, Poole–Frenkel

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(Some figures may appear in colour only in the online journal)

1. Introduction

Thermally stable phthalocyanine macrocylic compounds are known to exhibit electrical semiconductivity properties, stimulating active fundamental research interest for a wide range of applications in organic and flexible electronics [1]. Lanthanide bis-phthalocyanine sandwich compounds are

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characterised by their free radical character which facilitates electron transfer from one macrocycle to another due to overlap of π -orbitals [2]. Unsubstituted lutetium bis-phthalocyanine (LuPc₂) is believed to be the first intrinsic molecular semiconductor with room temperature (rt) steady state conductivity of $6 \times 10^{-2} \, \mathrm{S} \, \mathrm{m}^{-1}$ [3, 4]. Substituted liquid crystalline LuPc₂ molecules with eight long alkyl chains exhibit ordered liquid crystalline mesomorphic properties over temperature range between 100 and 500 K and the carriers hopping between the columns are found to be responsible for the frequency conductivity between 10^{-3} and $10^{5} \, \mathrm{Hz}$ [5]. A drift mobility of $2 \, \mathrm{cm}^2 \, \mathrm{V}^{-1} \, \mathrm{s}^{-1}$ is reported from in-plane conduction through evaporated LuPc₂ thin films between two gold electrodes [6]. The localisation of π -electrons on one of its two macrocycles produces The Heisenberg

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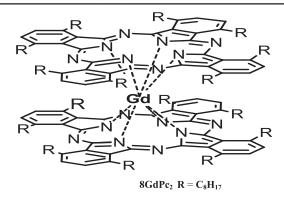


Figure 1. Chemical structure of non-peripherally octaoctyl-substituted bis-1,4,8,11,15,18,22,25-octakis(octyl) phthalocyaninato gadolinium.

type paramagnetic behaviour of LuPc₂ molecules [7]. Following the initial research carried out on LuPc₂ compounds, other rare-earth bisphthalocyaninate compounds such as dysprosium (Dy), europium(Eu), gadolinium (Gd), praseodymium (Pr), terbium (Tb), ytterbium(Yb) have since been investigated for their uses in resistive, optical, electrochemical, impedance or mass sensors [8–10].

Recent applications of plastic electronics are based on those derivatives bearing substituents that confer both solubility in organic spreading solvents and columnar liquid crystal behaviour typically at elevated temperatures. The conductivity of spin-coated thin films of octa(13,17-dioxanonacosane-15-sulfanyl)-substituted mesomorphic LuPc2 is found to decrease by four orders of magnitude on annealing to 140 °C for 3 h due to formation of a staggered slipped stacking structure [11]. Anthracene doping of up to wt 5% into spin-coated films of octakis(alkylthio) substituted Lu bisphthalocyanine is reported in recent years to increase the conductivity by two orders of magnitude with simultaneous decreases in activation energy [12]. Spin-coated films, ~40 nm thick, of alkyl-substituted dysprosium phthalocyanine molecules showed an increase in rt Ohmic conductivity by two orders of magnitude from 6.57×10^{-8} to $6.42 \times 10^{-6} \,\mathrm{S\,m^{-1}}$ as the sample was annealed at the liquid crystalline temperature of 350 K, implying the formation of thermally induced ordered film [13]. $2.12 \times 10^{-7} \, \text{m}^2 \, \text{V}^{-1} \, \text{s}^{-1}, \quad 6.72 \times 10^{-7} \, \text{m}^2 \, \text{V}$ $6.72 \times 10^{-7} \,\mathrm{m}^2 \,\mathrm{V}^{-1} \,\mathrm{s}^{-1}$ $21.58 \times 10^{-6} \,\mathrm{m}^2 \,\mathrm{V}^{-1} \,\mathrm{s}^{-1}$ were estimated for hole mobility of spin-coated tetrasubstituted Lu, Eu, Yb thin films, respectively [14]. The Ohmic conductivity of 5 nm thick thermally deposited TbPc2 under high vacuum conditions is found to be $0.032\,\mathrm{S\,m^{-1}}$ at 303 K with the value of $0.158\,\mathrm{eV}$ for the temperature independent activation energy, implying the thermal excitation of carriers from a continuous density of deep trap states [15].

This paper reports for the first time bulk ambipolar charge transport in 5 μ m thick solution processed films of a novel liquid crystalline gadolinium bis-phthalocyanine complex, 8GdPc₂ in figure 1, bearing a total of 16 octyl chains (R = C₈H₁₇) as substituents on its non-peripheral positions. Thermally induced phase changes were investigated from Differential Scanning Calorimetry (DSC) and UV-vis

absorption spectra for bulk materials and spin-coated films, respectively. The time of flight (TOF) technique was employed to determine values of electron and hole mobilities $\mu_{\rm e}$ and $\mu_{\rm h}$ at temperatures below and above the crystal to mesophase transition. This method involved the measurement of the time required for a sheet of charge carriers photogenerated near one of the electrodes by pulsed light irradiation to drift across the sample to the other electrode under an applied electric field. In this way it is possible to study the fastest charge transport and recombination in organic 8GdPc₂ semiconductors involving the mechanisms within molecules, between molecules, as well as between crystal planes and grains [16]. Similarly substituted metal free phthalocyanine molecules (8H₂Pc) are found to exhibit the unipolar charge transport behaviour with the hole ${\approx}10^{\mbox{--}4}\,\mbox{m}^2\,\mbox{V}^{-1}\,\mbox{s}^{-1}$ in the \mbox{Col}_h and \mbox{Col}_r phases. This value is two orders magnitude higher than one obtained in the present case [17].

2. Experimental methods

Synthetic routes to bis-complexes of tetrapyrrole based macrocycles containing a lanthanide metal ion are well established [18]. The metal free 1,4,8,11,15,18,22,25-octakis (octyl)phthalocyanine was reacted with gadolinium acetate under reflux in octanol with 1,8-Diazabicyclo[5.4.0]undec-7ene (DBU) as a promoter. DSC curves were recorded for asprepared bulk material 8GdPc2 over a temperature range between -50 °C and 200 °C using a Linkham THM600 hot stage, and with a TA Instruments DSC 10 instrument coupled to a TA200 workstation. A polarised light optical microscope (Olympus BH2 polarising microscope with a Linkam THM600 hot stage was employed to observe corresponding mesophase textures of the molecule. Using a Hitachi U-3000 ultraviolet-visible (UV-vis) spectrometer fitted with a Mettler FP80 processor coupled to a Mettler FP82 hot stage, optical absorption spectra for 8GdPc2 spun thin films on ultrasonically cleaned quartz substrates were recorded in the 300-900 nm at different temperatures to investigate temperature dependent molecular reorganisations. Spin coated films of 8GdPc₂ were prepared by conventional methods using a KW-4A spin coater from the Chemat Technology Inc., USA operating at a rotation speed of 1500 rpm for 30 s. The spreading solution (10 mg ml⁻¹) of the 8GdPc₂ compound in chloroform (99.9% anhydrous) was used.

The TOF measurements were performed for determining drift mobilities in a 5 μ m thick drop-cast film sandwiched between two transparent indium tin oxide (ITO) coated glass substrates. A DekTak profilometer was employed to measure the film thickness. A liquid crystal cell was placed in a modified Linkam LTS350 hot stage filled with 8GdPc₂ by heating at 185 °C, i.e. above the mesophase to isotropic liquid transition temperature. The sample was then cooled at a rate of 2 °C min⁻¹ to help promote larger crystal formation. Antiparallel Polyamide was used as the alignment agent between the organic layer and the ITO electrodes. The photocurrents were produced using a 532 nm pulsed output of a

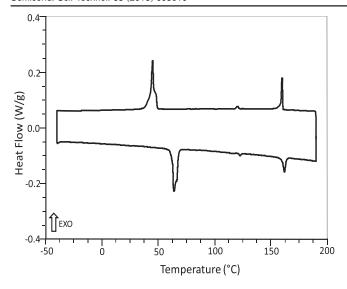


Figure 2. Differential scanning calorimetric curves for 8GdPc₂ for heating and cooling cycles.

double-frequency Nd:YAG laser. The top and bottom electrodes were connected to an external circuit via a variable resistor and a DC power supply which provided the bias. The resulting photocurrent transient was recorded on a high-resolution digitising oscilloscope. For the measurement of electron transport, the polarity of the power supply was reversed. Signal averaging and background subtraction were carried out on all signals to improve data quality. Further details on the instrumentation are available from our recent publication [19].

3. Results and discussion

The DSC curve in figure 2 shows structural transitions that occur on the second heating/cooling cycle at temperature T with enthalpy changes shown in brackets: 64.2 °C $(19.4 \,\mathrm{J}\,\mathrm{g}^{-1}), 121.6\,^{\circ}\mathrm{C} (0.41 \,\mathrm{J}\,\mathrm{g}^{-1}), 162.2\,^{\circ}\mathrm{C} (3.9 \,\mathrm{J}\,\mathrm{g}^{-1})$ (Heating-2nd cycle), $160.6 \,^{\circ}\text{C}$ (5.1 J g⁻¹), 121.6 °C $(0.56 \,\mathrm{J}\,\mathrm{g}^{-1})$, $46.4\,^{\circ}\mathrm{C}$ $(18.9 \,\mathrm{J}\,\mathrm{g}^{-1})$. The sharp peak on the heating cycle (lower line) of DSC curve at 64.2 °C is assigned to the transition of the crystal (K) to rectangularly packed columnar mesophase (Col_r) transition. The small peak, low enthalpy change, at 121.6 °C implies a mesophase to mesophase $(Col_r \rightarrow Col_h)$ transition where h refers to hexagonal packing and that at 162 °C corresponds to melting into the isotropic liquid phase (I). Mesophase types were assigned on the basis of well documented characteristic birefringence textures [20]. It is observed from optical pictures in figure 3 that liquid crystal textures were formed sequentially as the sample was heated. The transition from the columnar rectangular mesophase (Col_r) to the columnar hexagonal mesophase (Col_b) was observed. Peripherally octaoctyl-substituted gadolinium phthalocyanine derivatives exhibited similar phase transition from K to Col_r at 61 °C while $Col_r \rightarrow Col_h$ transition is associated with the temperature of 93 °C [21]. However, the thermal behaviour of non-peripherally substituted Pcs is largely determined by the interaction between substituent chains on neighbouring benzo-moieties. The re-arrangement of molecular packing takes place within the bulk structure of films of these compounds on thermal annealing leading to enhanced electrical characteristics [22].

UV-vis spectra of the 8GdPc₂ film in figure 4 show Soret and Q-bands in the range between 340 nm and 360 nm, and 640 nm and 730 nm, respectively due to long range face-toface stacking arrangements of the monomers. The absorption at 490 nm is generally associated with the free radical structure of a typical bis-phthalocyanine sandwich compounds. The broad band at 318 nm is assigned to the B band $(\pi - \pi^*)$ absorption and it is one of the characteristics absorption bands for bis phthalocyanine. According to the extended Hückel molecular orbital model, doubly degenerate lowest unoccupied molecular orbital (LUMO) is believed to have centred on the pyrrole and isoindole nitrogen. The interactions between the macrocyclic rings split the π highest occupied molecular orbital (HOMO) levels in the lanthanide sandwich complexes [23]. The relative energy distance between the bonding HOMO and LUMO levels is estimated to be 1.73 eV. The radical band at 473 nm (2.62 eV) is the charge transfer band from the inner doubly degenerate orbital of e₁ to half-filled a₂ orbital. These optical transitions are schematically described in figure 4(b) in terms of energy levels corresponding to singly degenerate orbitals and doubly degenerate orbitals. The present spectral characteristics are broadly in keeping with ones observed for quasi-Langmuir-Shafer films of peripherally substituted bis(phthalocyaninato) holmium complexes [24]. The spectra at rt and 35 °C are identical, exhibiting the Q-band absorptions at 726 nm and 642 nm. At 70 °C, i.e. within the first mesophase (Col_r) range, the band at 726 nm has sharpened and exhibits an increase in intensity. The band at 642 nm shifts to 640 nm, showing a smaller increase in intensity. At 130 °C, corresponding to the second mesophase range (Col_h), the band at 726 nm has undergone a small decrease in intensity with the 640 nm band returning to 642 nm.

Figure 5(a) shows hole photocurrents on double logarithmic scales at 30 °C, 70 °C and 130 °C for the bias of $V_a = +30 \,\mathrm{V}$. The transit time t_0 of photo-generated carriers corresponding to the respective inflection points determined by the intersection of the tangents to the initial and post flight parts of the curves and clear inflection points, t_0 , can be observed at 3.18, 5.33 and 1.76 µs corresponding to the transition temperature of 30°C, 70°C and 130°C, respectively. The traps for two types of carriers can be distinguished by changing the polarity of bias [25]. Therefore, electron transport was also investigated in a single experiment for the same sample for reverse bias of $V_a = -30 \,\mathrm{V}$ and values of 26.5, 17.80 and 5.21 μ s were obtained for t_0 at the corresponding temperatures from figure 5(b). The primarily dispersive nature of transport for both types of carriers is consistent with the multidomain film structures which consist of many boundaries. The measurements at rt were repeated and values of $\mu_{(e)}$ and $\mu_{(h)}$ were estimated at temperature T from the knowledge

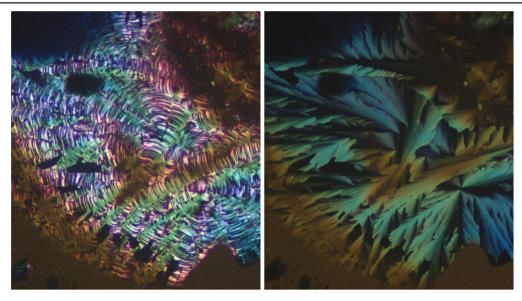


Figure 3. Polarised optical microscopic texture at room temperature after annealing from the isotropic liquid.

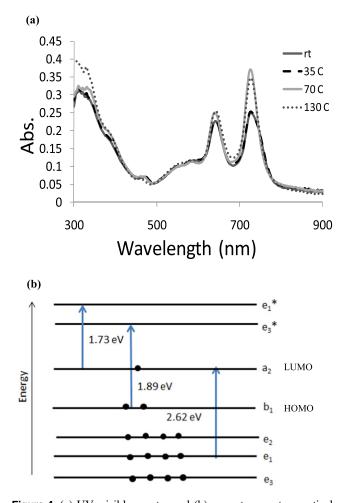


Figure 4. (a) UV–visible spectra and (b) room temperature optical transitions between orbitals $(\mathbf{a_2} \rightarrow \mathbf{e_1}^*)$, $(\mathbf{b_1} \rightarrow \mathbf{e_3}^*)$ and $(\mathbf{e_1} \rightarrow \mathbf{a_2})$ correspoding to 730 nm band, 764 nm band and radical band of a spin-coated 8GdPc₂ film.

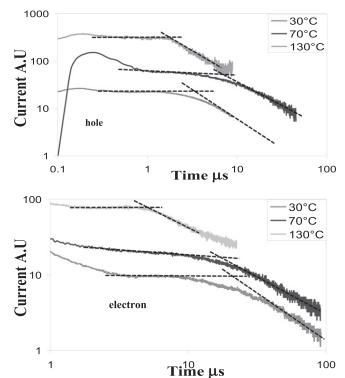


Figure 5. Hole and electron photocurrent transients, measured at room temperature in a 5 μ m thick cell of 8GdPc₂ for K, Col_r and Col_h mesophases corresponding to transition temperature of 30 °C, 70 °C and 130 °C.

that

$$\mu_{(e,h)}(T, E) = \frac{d^2}{t_0 V_a},$$
 (1)

where the film thickness $d = 5 \mu \text{m}$, electric field $E = \frac{V_a}{d}$.

The measurements were repeated at room temperature and the mean value of the results of the calculations have been summarised in table 1. Both electron and hole mobilities are critically dependent of the mesophase of the $8GdPc_2$ films. The mobilities of both carriers in Col_h mesophases are found to be higher by factor of ≈ 3 than those in Col_r mesophases. TOF measurements were performed on solution processed liquid crystalline metal free phthalocyanine molecules with octahexyl (C_6H_{13}) substitution on similar non-peripheral positions, showing similar mesophase behaviour. High drift motilities of up to $1.4~cm^2~V^{-1}~s^{-1}$ and $0.5~cm^2~V^{-1}~s^{-1}$ for both hole and electrons in the crystalline solid phase of ($6PcH_2$) were obtained at rt from the TOF measurement [26]. The mobility values of $8GdPc_2$ are at least one order of magnitude higher than those reported for non-peripherally octahexyl substituted ambipolar zinc ($ZnPc_6$) phthalocyanine [27].

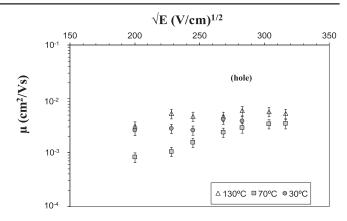
Recently reported values of $4 \times 10^{-9} \, \text{m}^2 \, \text{V}^{-1} \, \text{s}^{-1}$ and $8 \times 10^{-10} \,\mathrm{m^2 \, V^{-1} \, s^{-1}}$ for the field effect electron and hole mobilities, respectively of ambipolar Gd-bisphthalocyanine active layer vacuum sublimed on silicon substrates are significantly smaller than those obtained from the present TOF measurements [28]. Similarly the holes and electrons mobility of $0.11 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for and $0.06 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, respectively were obtained for the solution processed quasi-Langmuir-Shäfer film of homoleptic sandwich-type [2,3,9,10,16,17,23,24-octa(naphthoxy)phthalocyaninato] europium triple-decker complex as an active layer in the organic field effect transistor (OFET) [29]. However, the carriers are often injected from the contacts into the channels of OFET transistors and values of OFET mobilities are likely to be limited because of the injection of carries from the channel into the [30]. Values 0.15–0.35 cm² V⁻¹ s⁻¹ for one-dimensional intracolumnar carrier mobility were reported for liquid crystalline thioalkylated Lu, Eu, Tb bisphthalocyanine using pulse-radiolysis time-resolved microwave conductivity (PR-TRMC) technique [31]. The PR-TRMC usually gives larger mobility than TOF because the latter technique involves the transport of charge carriers over relatively large distances in multidomain structures encountering the traps of long life [32]. In view of these considerations, the present TOF mobilities represent the realistic ambipolar behaviour of 8GdPc2 films with potential device applications.

Measurements were repeated between $10V \leqslant V_a \leqslant 35V$ with a view to examining the effect of applied field on the mobility. The mobilities of both types of carriers are shown in figure 6 as function of \sqrt{E} . The linear relation indicates the Poole–Frankel type field dependence of the mobility at temperature T in the form;

$$\mu_{(e,h)}(T, E) = \mu_{(e,h)}(T, E = 0) \exp(\beta \sqrt{E}).$$
 (2)

The errors in $\mu_{(e,h)}$ primarily arise from the dependence of the electric field E on thickness d. The contributions of the transit time to error bars are significantly small. The least square fitting of equation (2) to experimental data is also included in the error bars.

Values of the field lowering coefficients β are estimated in the order of $+10^{-5}\,\mathrm{m}^{5/2}\,\mathrm{V}^{-3}\,\mathrm{s}^{-1}$ for holes and $+10^{-6}\,\mathrm{m}^{5/2}\,\mathrm{V}^{-3}\,\mathrm{s}^{-1}$ for electrons. The physical interpretation



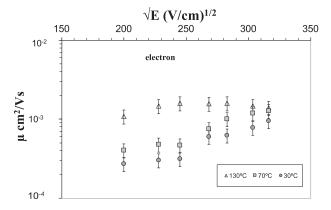


Figure 6. Poole–Frankel plots for hole and hole mobilities in 8GdPc₂ for K (circle), Col_r (rectangle) and Col_h (triangle) mesophases corresponding to transition temperature of 30 °C, 70 °C and 130 °C, respectively.

of these positive values of β may be obtained by examining their dependence on the Gaussian density of energy states of width σ and the dimensionless parameter defining the positional disorder Γ through the following form:

$$\beta = 0.78 \sqrt{\frac{qR}{\sigma}} \left[(\sigma/kT)^{3/2} - \Gamma \right], \tag{3}$$

where R is the intersite spacing parameter [33]. It is obvious from equation (3) that the positional order Γ is small in comparison with energy states of width σ implying the liquid crystalline $GdPc_2$ films are well structured. Similar positive field dependent mobility behaviour was observed for liquid crystalline semiconducting polymers based on poly(2,5-bis(3-alkylthiophen-2-yl)thieno[3,2-b]thiophene) [34]. The energy state of width for holes in Col_h mesophases is estimated to be 70% narrower than those in Col_r mesophases from the intercept at E=0 of the Poole–Frankel plots using equation (4) in the following form:

$$\mu_{(e,h)}(T, E = 0) = \mu_{0(e,h)} \exp\left[-\left(\frac{3}{5} \frac{\sigma}{kT}\right)^2\right],$$
 (4)

where $\mu_{0(e,h)}$ is the carrier mobility in the energetically disorder-free system at E=0. Similar calculations have been repeated for electrons and it is found that the energy states for Col_r mesophases are wider than Col_h mesophases by more than 200%.

Holes Electrons t_0 t_0 $\mu_{(h)}$ (10⁻⁶ m² V $\mu_{(e)}$ $^{-1}$ s⁻¹) (μs) (μs) $(10^{-6} \, \text{m}^2 \, \text{V})$ Mesophase Transition temperature (°C) K 30 3.18 0.262 2.65 0.0341 Col_r 70 5.33 0.156 0.0468 17.8 130 1.76 0.473 5.21 0.159 Col_h

Table 1. Summary of mobility calculations using equation (1).

4. Concluding remarks

The synthesis of solution processed liquid crystalline non-peripherally octaoctyl-substituted gadolinium bis-phthalocyanine complex, 8GdPc₂ ambipolar organic semiconductors is important for development of organic complementary metal oxide semiconductor (CMOS) circuits and organic light-emitting transistors. Transition temperatures for phase changes have been well defined by DSC curves and UV-visible absorption spectra. The ambipolar charge transport in 8GdPc₂ compounds is consistent with relative positions of HOMO and LUMO levels derived from optical transitions. The TOF mobilities of both types of carriers which can be selectively tuned by the annealing temperature of the 8GdPc₂ films are bulk characteristics and therefore these properties may be suitably exploited for design and fabrication of all organic complimentary printable circuits.

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