

Wheat Straw Pre-treatments using Eco-Friendly Strategies for Enhancing the Tensile Properties of Bio-based Polylactic Acid Composites

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Abstract

Individual and hybrid pre-treatments, including hot water, steam and microwave were used on wheat straw to mitigate the surface quality deficiencies for an intimate interfacial bond between substrate and Polylactic Acid matrix. Microstructural analysis showed the expansion of cells due to the pre-treatments which in turn promotes mechanical entanglement between wheat straw and PLA, thus, improving the tensile properties of bio-based composites. The tensile strength, elastic modulus, and toughness increased by 166%, 68%, and 285%, respectively, for pre-treated wheat straw compared to untreated.

1. Introduction

The utilisation of straw biomass in bio-based composites is gaining momentum due to their cost efficiency, lightweight, low density and less environmental impact during production (Mati-Baouche et al., 2014; Sahai and Pardeshi, 2019). So far, the most commonly used material for bio-based composites fabrication is wood (Anugwom et al., 2019; Kaboorani, 2017) but, wheat straw as a renewable material has the potential to successfully replace wood in various applications. In general, wheat straw has similar chemical constituents to those of wood (i.e. cellulose, hemicellulose, lignin, and extractives). However, wheat straw has a higher content of hydrophobic waxy cuticle layers and high amount of inorganic silica and extractives (Alemdar and Sain, 2008a; Ghaffar and Fan, 2015a, 2015b, 2014, 2013; Han et al., 2009). The comparatively poor surface properties of wheat straw with hydrophobic components can reduce the interfacial bond quality between the substrate and polymer binder. This has eventually hampered the commercialisation of wheat straw bio-based composites (Ghaffar et al., 2017a, 2017b; Ghaffar and Fan, 2017). Several other lingo-cellulosic biomass, including cereal straw, hemp and jute strands, henequen strands, corn cobs, cotton stalks, kenaf, bamboo and rice husks, sunflower stalks, and Bagasse have been proposed as reinforcements for bio-based composite fabrication (English and Spelter, 1992; Guo et al., 2018; Liu et al., 2017; Melo et al., 2014; Tarrés et al., 2019; Tran et al., 2014; Tribot et al., 2018; Yu et al., 2019).

35 For instance, Scaffaro et al. reported that the incorporation of reinforcing agents such as Posidonia
36 Oceanica leaves and Opuntia Ficus Indica in poly-lactic acid (PLA) and MaterBi® (MB) polymeric
37 matrices increase the stiffness, while decreasing toughness, elongation at break and tensile strength
38 of bio-based composites (Scaffaro et al., 2020, 2019, 2018).

39 Several methods have been developed in order to improve the surface compatibility of the natural
40 reinforcements by additive incorporation as well as employing a variety of pre-treatment processes
41 to produce a high-performance bio-based composites. Physical pre-treatment methods, including
42 steam cooking, steam explosion, liquid-plasma, and ozonation (Ghaffar et al., 2017b; Putra et al.,
43 2020), as well as chemical pre-treatments such as mercerization, acetylation, enzymatic, and alkaline
44 pre-treatments (Ghaffar and Fan, 2015a; Hýsek et al., 2018; Hýsková et al., 2020; X. Li et al., 2007),
45 have been shown to be beneficial in terms of (i) modifying the surface of straw with the partial removal
46 of extractives, waxes, and silica which made it more hydrophilic and more compatible with water-
47 based resins, (ii) expanding the microcellular structure of straw and hence improving the penetration
48 of matrix into cell lumens, leading to intimate bonding quality between substrate and matrices
49 (Elbashiry et al., 2018; Oushabi, 2019). Wang et al. (2018) employed an alkaline pre-treatment using
50 4 wt % sodium hydroxide (NaOH) solution for bamboo fibers, which led to an increase in the elongation
51 at break and tensile strength of the polylactide composites filled with alkaline treated fibres by 84%
52 and 49%, respectively (Wang et al., 2018). The chemical pre-treatment decreases the hydrophilicity of
53 biomass and increases the fibre surface's roughness, which increases the adhesion between fibres and
54 the surrounding polymeric matrices (Manshor et al., 2014). On the other hand, Santos et al. reported
55 the remarkable effectiveness of hot water pre-treatment of piassava *Attalea funifera* fibres which
56 increases the cellulose peak intensity and crystallinity index of the fibres (Santos et al., 2018). It should
57 be noticed that chemical pre-treatments are shown to be effective in the destruction and extraction
58 of a certain amount of lignin, hemicellulose, and pectin covering the biomass surface, resulting in
59 better-purified cellulose. However, the low cost-efficiency of the procedure in large scale production
60 and their harmful environmental impacts limits their extensive applications over physical pre-
61 treatments (Tian et al., 2018). Additionally, the use of coupling agents has proved to be effective,
62 especially with wood-based plastic composites (Ghaffar et al., 2018). Coupling agents such as Silane
63 (Gurunathan et al., 2015), maleic anhydride (MA) (Mishra et al., 2000), polydopamine (PDA) (Fan et
64 al., 2018), and attapulgitic clay (ATP) (Zhu et al., 2019) can modify the surface properties and enhance
65 the bonding/adhesion between cellulosic biomass and polymer matrices. The mechanism is formed
66 through the reaction between hydroxyl groups at one end, and natural fibre surface and another fun-
67 ctional group at the other end in order to bond with the polymer matrices (Ghaffar et al., 2018).

68 In this study, as a novelty, wheat straw was optimised by selective separation of defects in the context
69 of bio-based composites (i.e. nodes, see **Figure 1a**). Node has a very different microstructure to that
70 of the internode (Ghaffar and Fan, 2015a). It is much less structurally competent, specifically in
71 tension, furthermore, node's surface chemical functional groups, precisely their higher extractives and
72 hydrophobic wax, inhibit the good interfacial bonding between the substrate and matrix in bio-based
73 composites (Ghaffar et al., 2017a; Ghaffar and Fan, 2015a). Therefore, after the separation of nodes,
74 the internodes of wheat straw were subjected to various hybrid pre-treatments, including hot water
75 followed by steam for two different time durations and microwave pre-treatment. **The employed pre-**
76 **treatments are cost-effective and prevent the entry of hazardous and toxic chemical reagents to the**
77 **environment.** Comprehensive materials characterisation were conducted, and bulk property
78 evaluation of bio-based composites was also carried out in order to evaluate the effects of pre-
79 treatments on potential tensile strength improvements.

80 **2. Materials and methodology**

81 As-received material, wheat straw (*Triticum aestivum L.*), was obtained from a residential farm,
82 Middlesex, United Kingdom, harvested in late summer of 2019. The material was baled and dried
83 directly on-site. The thermoplastic polylactic acid (PLA), supplied by Tecnar GmbH, Ilsfeld, Germany,
84 was used as the matrix binder in this study. All materials were stored in ambient room temperature
85 to keep air-dryness.

86 **2.1 Wheat straw preparation for pre-treatment**

87 Prior to the pre-treatment stage, as-received straw was bailed out and prepared for the cleaning stage.
88 As previously reported (Harper and Lynch, 1981), straw is comprised of internodes ($57 \pm 10\%$), nodes
89 ($10 \pm 2\%$), leaves ($18 \pm 3\%$), chaffs ($9 \pm 4\%$) and rachis ($6 \pm 2\%$) by mass, which leads to exhibit an
90 inconsistency in chemical distribution. The wheat straw stems were cut above and below the nodes
91 for the internode separation, and the internode parts were extracted by hand. After the cleaning
92 process, all samples were dried in the oven at 100 ± 5 °C for 24 hours. In total, three hazardous-
93 substance-free pre-treatments with straw to water ratio of 1:100 (by weight) and starting water
94 temperature of 100 °C were conducted in this study (see **Figure 1 b-e**).

95 **Figure 1**

96 **2.1.1 H+S pre-treatment**

97 H+S pre-treatment contains two consecutive stages (i.e. hot water and steam stages). In the first stage,
98 the straw was introduced into a pressure cooker for 60 minutes, which was tightly sealed to maintain
99 a constant pressure of approximately 0.1 MPa. Hot water (H) stage was then followed by steam
100 treatment stage (S) by removing the straw from hot water and placing them into mesh basket

101 instantly, which was positioned directly above boiling water inside the pressure cooker. In this stage,
102 the steam was allowed to pass through the wheat straw for another 30 minutes.

103 **2.1.2 0.5H+S pre-treatment**

104 The 0.5H+S pre-treatment was conducted exactly as the H+S pre-treatment procedure, except the hot
105 water stage, was shortened to 30 minutes (i.e. 0.5H).

106 **2.1.3 H+M pre-treatment**

107 The term H refers to the first pre-treatment stage (i.e. Hot water) in which the straw was boiled in the
108 pressure cooker for 60 minutes. While the term M refers to microwave pre-treatment in which the
109 straws were spread evenly on the tray and placed into standard microwave at 900 watts for 15
110 minutes.

111 **2.2 Experimental plan**

112 **Figure 2** illustrates the testing programme followed for this study.

113 **Figure 2**

114 **2.3 Bio-based composites sample preparation**

115 Before the composite production, all the untreated and pre-treated straws were cut into 5-15 mm
116 pieces by employing Retsch SM 100 cutting mill for bio-based composites. **The composites in this study**
117 **were fabricated using PLA and internode part extracted from wheat straw with a ratio of 1:1 by weight.**

118 For each sample, the chopped straw and polymer pellets were mixed to achieve the uniform
119 arrangement of material (see **Figure 3a**). The mixture was placed into the steel mould of size 100mm
120 x 100mm x 20mm (see **Figure 3b**). The mould was then placed in the hot-air oven for 15 minutes at
121 200 ± 10 °C to facilitate the PLA melting process. After the pre-heating step, the mould with softened
122 PLA and straws was hot-pressed for 20 minutes at 180 °C under the pressure of 10 MPa. After
123 removing the mould from the hot-press, a mass of 15 kg was placed on top of the samples to prevent
124 the expansion phenomenon. Furthermore, the manufactured composite was left for 30 minutes to
125 cool down to reach the ambient temperature. Finally, the finished products were cut precisely into
126 20mm wide strips to form tensile test samples. Four stripes for tensile strength evaluation were cut
127 out from one strawboard (see **Figure 3c**).

128 **Figure 3**

129 **2.4 Characterisations and testing**

130 **2.4.1 Microstructure**

131 The morphology of wheat straw was assessed by scanning electron microscopy (SEM). Cross-section
132 of the untreated and pre-treated internodes were studied using Supra 35VP, Carl Zeiss, Germany. To
133 obtain a reliable statement about the microstructure and the changes induced as a result of pre-
134 treatments, at least ten samples with the size of 10 mm³ were analysed. ImageJ software was

135 employed in order to quantitatively evaluate the thickness of the epidermis layer and area percentage
136 of the parenchyma.

137 **2.4.2 Crystallinity**

138 The powder X-ray diffraction was performed to investigate the changes in the crystalline structure of
139 four powdered samples of untreated, H + S, 0.5H + S, and H + M employing a D8 advanced Bruker AXS
140 diffractometer, Cu-K α radiation, wavelength of 1.542 Å, 2 θ of 20-40° at 40kV and 40 mA.

141 Based on the XRD results, the crystallinity index (CI) was calculated by an empirical method suggested
142 by Segal et al. (1959) (**Eq.1**) (Segal et al., 1959). However, the CI term is reliable only in the interest of
143 comparison as it specifies the order of crystallinity rather than the crystallinity of crystalline regions.

$$144 \quad CI (\%) = \frac{I_{002} - I_{amorph}}{I_{002}} \times 100 \quad (1)$$

145 In this equation, the I_{002} is the maximum intensity of the (0 0 2) lattice diffraction and represents both
146 crystalline and amorphous region of the materials (i.e. 2 θ between 20° and 24°), and I_{amorph} represents
147 the amorphous region of the material (i.e. 2 θ between 17° and 20°).

148 **2.4.3 Surface chemical functional group distribution**

149 Attenuated total reflectance-Fourier transform infrared spectroscopy (FTIR-ATR) was used by
150 employing a PerkinElmer Spectrum one Spectrometer. The samples, including the inner and outer
151 surface of randomly selected untreated and pre-treated straw internodes, were mounted on an (ATR)
152 equipped with a 3x bounce diamond crystal with an incident angle of 45°. The instrument produced
153 20 scans with a resolution of 4 cm⁻¹ at a wavenumber ranged between 4000 cm⁻¹ and 600 cm⁻¹. A batch
154 of three samples have been tested for each composition and the average results are reported.

155 **2.4.4 Water contact angle**

156 The water contact angle test was performed to assess the wettability of wheat straw outer surface
157 using the First Ten Angstroms FTA 1000 Analyser System. The angle between the water droplet surface
158 and the sample surface was measured every 100 s for a total of 500 s. Ten different specimens were
159 tested for each pre-treatment. The wettability of each sample was investigated by measuring the
160 initial contact angle (θ_i), and the equilibrium contact angle (θ_e). The penetration and spreading
161 constant (K) values (Shi and Gardner, 2001), were calculated by employing the wetting model (i.e. **Eq.**
162 **2**) in OriginPro software. In principle, K value determines the liquid penetration proficiency within the
163 surface of wheat straw.

$$164 \quad \theta = \frac{\theta_i \theta_e}{\theta_i + (\theta_e - \theta_i) \exp \left[K \left(\frac{\theta_e}{\theta_e - \theta_i} \right) t \right]} \quad (2)$$

165 **2.4.5 Thermogravimetric analysis**

166 The thermogravimetric analysis of straw was performed to compare their degradation characteristics.
167 The thermal stability of each sample was obtained employing an SDT Q600 V8.3 Build 101 (TA

168 instrument). The samples of about 3-4 mg were heated from 20 to 650 °C at a rate of 10 °C/min in a
169 continuous flow of nitrogen (i.e. nitrogen atmosphere) to avoid unwanted oxidation.

170 **2.5 Mechanical properties**

171 The tensile strength performance of the samples was assessed according to the ASTM D3039/D3039M
172 using Instron 5969 universal testing machine equipped with wedge action tensile grips. The tensile
173 tests were carried out on the strawboard strips of size 100mm x 20mm x 20mm. Specimens were
174 inserted 40mm into the grips to create a 20mm gauge length. A total of 20 samples were tested, and
175 relevant data collected. Modulus of elasticity, elongation at break, and tensile toughness were also
176 calculated from the stress-strain curve by measuring the slope at the elastic region, percentage of
177 maximum extension vs. initial size of the sample at the point of rupture, and integrating the area
178 under the stress-strain curve, respectively, employing the tangent and integration gadgets in OriginPro
179 software.

180 **2.6 Physical properties**

181 The dimension stability of the strawboards, including thickness swelling (TS) and water absorption
182 (WA) tests, was evaluated in accordance with the BS 5669-1:1989. Prior to the tests, the thickness
183 and weight of each specimen were precisely measured by means of digital equipment. Six samples
184 for each specimen type were tested.

185 **3. Results and Discussion**

186 **3.1 Microstructural changes due to pre-treatments**

187 In order to analyse and compare the microstructure of untreated and pre-treated wheat straw, SEM
188 was used to investigate the cross-section of the wheat straw internode. Results revealed that all the
189 pre-treatments, induced the desired changes in the microstructure of wheat straw internode,
190 precisely, reducing the thickness of epidermis as a result of hot water (H) stage and expansion of
191 parenchyma, which could potentially facilitate the deeper and more efficient penetration of polymer
192 matrix in liquid form. The aforementioned desired changes lead to better quality bonding and
193 subsequently, higher performing bio-based composites. **Figure 4 a-d** shows that the pre-treatment
194 reduced the thickness of the epidermis layer from 103.6 µm for the untreated straw to 54.8 µm, 65.7
195 µm, and 70.9 µm for H+S, 0.5H+S, and H+M pre-treated samples, respectively. Since the epidermis
196 layer contains hydrophobic silicone and waxes components, reducing the size of this layer could be
197 beneficial as it means the polymer matrix will bond better with the wheat straw due to the reduced
198 hydrophobic contact area. In addition, the area percentage of the parenchyma for each sample was
199 calculated by employing the ImageJ software. The results revealed that H+S and 0.5H+S pre-
200 treatments, i.e. **Figure 4 b and c**, show a noticeable expansion of parenchyma, approximately 12% for
201 H+S and 10% for 0.5H+S compared to untreated wheat straw samples, therefore, becoming broader

202 and more in-depth. Moreover, the thickness of the cell walls, which can affect the mechanical
203 interlocking with polymeric matrices, slightly decreased. It is evident that the steaming (S) stage of H
204 + S and 0.5H + S pre-treatments has led to the deepening of wheat straw layers, which resulted in the
205 expansion of parenchyma. The same modifications can be observed in H + M pre-treatment (**Figure 4**
206 **d**) without the steaming stage, i.e. 9% expansion compared to untreated sample. This reflects the
207 microwave (M) stage of the pre-treatment acting as a pressure steaming. Kashaninejad et al. (2010)
208 also reported that the microwave pre-treatment increased the internal surface area of the wheat
209 straw structure, which was evidenced by a decrease in bulk and particle density as well as an increase
210 in the overall particle size of wheat straw (Kashaninejad et al., 2010).

211 **Figure 4**

212 **3.2 Physicochemical changes due to pre-treatments**

213 The FTIR-ATR spectroscopy was employed to characterise the chemical distribution of the inner and
214 outer surface of straw internodes. The node parts of the straw stem are separated and taken out from
215 the experiments following the authors' previous work, which indicated that nodes possess higher
216 levels of hydrophobic wax and silica on their surface compared to the internodes (Ghaffar et al., 2017a;
217 Ghaffar and Fan, 2015b). This was proven to be an inhibiting factor for interfacial bonding quality in
218 the context of the composite (Ghaffar et al., 2017a). The bands at 790 cm^{-1} and 985 cm^{-1} are referred
219 to the silicon carbide (Si-C) and silicon oxide (Si-O) stretching vibration bonds, respectively (Frost and
220 Mendelovici, 2006), which presented only in the spectrums obtained for the outer surface of wheat
221 straw internode (see **Figure 5 b**). This suggested that silicon components mainly appeared on the outer
222 surface. Pre-treatments reduced silicon components, particularly in H + S pre-treatment, there is a
223 drastic reduction in the silica bands on the outer surface.

224 **Figure 5**

225 In order to fully validate the silicon components' reduction phenomenon, the thermal degradation of
226 both untreated and H+S pre-treated wheat straw internode has been assessed using TGA analysis,
227 shown in **Figure 6**. In the TGA curves, there are three distinct degradation stages. The first stage (i.e.
228 between $100\text{-}150\text{ }^{\circ}\text{C}$) relates to initial weight loss generated the moisture evaporation, and it is highly
229 associated with the initial moisture of samples. The second stage (i.e. between $250\text{-}350\text{ }^{\circ}\text{C}$) refers to
230 the severe weight loss formed by cellulose and hemicellulose degradation. Finally, the third stage is
231 representing the decomposition of non-cellulosic components (i.e. lignin) (Ghaffar and Fan, 2015a).
232 The enhanced thermal stability of straw and the high onset of degradation temperature are directly
233 linked together, meaning that the higher degradation onset signifies the better thermal stability of
234 straw (Ghaffar and Fan, 2015a; Zandi et al., 2019). The onset of degradation temperature for H+S pre-
235 treated straw (i.e. $315\text{ }^{\circ}\text{C}$) is relatively higher than that of untreated straw (i.e. $296\text{ }^{\circ}\text{C}$), which is due

236 to the extraction of hemicelluloses and waxy substances generated by the H+S pre-treatment.
237 Moreover, the higher crystalline structure of straw after H+S pre-treatment needs more temperature
238 for degradation concerning the untreated straw. This was confirmed by the intensity reductions at
239 bands 1735 cm⁻¹ for hemicelluloses extraction as well as 2920 cm⁻¹ and 2850 cm⁻¹ for the removal
240 of waxy substances (Saari et al., 2014). Moreover, the higher crystalline structure of straw after H+S
241 pre-treatment needs more temperature for degradation concerning the untreated straw (Alemdar
242 and Sain, 2008a).

243 On the other hand, the high volume of ash and silica on the outer surface of straw, negatively affects
244 the incorporation of these materials in bio-based composites applications. Qin et al. (2010) reported
245 that the Residual weight at 600 °C refers to the remained ash content (Qin et al., 2011). The results
246 revealed that the ash content after H+S pre-treatments is relatively lower (i.e. 4.5%) compared to the
247 untreated sample (i.e. 11%). The reduction in the ash and silica content of the straw improves the
248 interfacial bonding between the straw and polymer-matrices, which eventually leads to an increase in
249 the feasibility of using these materials in bio-based composites (Kaewtatip and Tanrattanakul, 2008).

250 **Figure 6**

251 Bands at 1595 cm⁻¹ and 1510 cm⁻¹ correspond to aromatic ring stretch of lignin, which is particularly
252 enhanced in the inner surface of pre-treated straw samples (Yang et al., 2020). The sharpest peaks of
253 lignin bands can be observed as a result of H + M and 0.5H + S pre-treatments, which suggest the
254 release and re-deposition of lignin on the surface of the straw during the pre-treatment process.
255 (Kaparaju and Felby, 2010; Kristensen et al., 2008; J. Li et al., 2007). The carboxyl groups in the acids
256 and esters of acetic, *p*-coumeric, ferulic, and uronic acids, was appeared at band 1735 cm⁻¹, which
257 represents the principal constituents of extractives and hemicellulose (Alemdar and Sain, 2008b). The
258 intensity of this band was reduced by H + S and 0.5H + S pre-treatments compared to H + M pre-
259 treatment, indicating higher removal of hemicellulose. Moreover, the absence of a 1735 cm⁻¹ peak in
260 the inner surface of wheat straw indicates that hemicellulose fractions are relatively lower in the inner
261 surface of the wheat straw. Sharp peaks at 2920 cm⁻¹ and 2850 cm⁻¹ both in outer and inner surfaces
262 represent asymmetric and symmetric CH₂ stretching bands, which correspond to the aliphatic
263 fractions of waxes (Merk et al., 1997). Results show that the outer surface of the internode consists
264 of higher amounts of wax and inorganic substances concerning the inner surface (see **Figure 5 a-b**).
265 The pre-treatments showed a positive effect in the reduction of inorganic substances and waxes. The
266 highest reduction in the intensity of this band occurred for H + S pre-treatment, which shows a notable
267 decrease in the amount of hydrophobic wax. The outer surface of the straw internode contains plenty
268 of cell elements, including parenchyma cells, fibres, vessel elements, and epidermis (Ghaffar et al.,
269 2017a). Among the aforementioned cell elements, the outmost surface cell (i.e. epidermis) comprises

270 dense and thick-walled cells covered with cutin cuticle wax. **Figure 7** demonstrates the water contact
271 angle results of untreated (UN) and H+S pre-treated straw samples. The results show the significant
272 impact of pre-treatment on the wettability of straw in which the θ_i and θ_e were decreased from 88.73°
273 and 69.79° for the untreated sample to 64.06° and 34.59° for H+S pre-treated samples, respectively.
274 As shown in **Figure 7**, the K value has been increased from 1.56×10^{-4} for the untreated sample to 10.5
275 $\times 10^{-4}$ for H+S pre-treatment. This shows that the H+S pre-treatment has led to hydrophilic surface
276 behaviour. The improvements obtained in the wettability of straw could be associated with two
277 phenomena. First, the H+S pre-treatment causes partial removal of some substances (i.e. lignin,
278 pectin, waxy substances, and natural oils), which cover the outer surfaces of straw stem (Mwaikambo
279 and Ansell, 2002). Second, the surface porosity improves as a result of extractives removal during hot
280 water (H) stage followed by expansion of pores during the steaming (S) stage, this, in turn, increases
281 the surface pores dimensions, leading to more hydrophilic surface behaviour.
282 According to several studies (Kaparaju and Felby, 2010; Mosier et al., 2005), the boiling stage can
283 result in partial hemicellulose extraction and changes to the lignin structure in the straw biomass. In
284 general, the reduction in the hemicellulose and cellulose content could be resulting in an increase in
285 porosity volume (Grethlein, H.E., 1985; Kaparaju and Felby, 2010; Zheng et al., 2018). The wettability
286 improvements generated by the H+S pre-treatment improves the surface matrices adhesion. Sun et
287 al. (2003) also reported a positive effect of hot water (i.e. approximately 80-90 C for 30 min) pre-
288 treatment of wheat straw, which released the original lipophilic extractives by 41-53% (Sun et al.,
289 2003).

290 **Figure 7**

291 **3.3 Crystallinity index changes due to pre-treatments**

292 Powder X-ray diffraction analysis of the wheat straw internodes was collected before and after pre-
293 treatments. As can be seen in **Figure 8**, the highest intensity was observed at $2\theta = 21.69-21.85^\circ$, which
294 represents the 002 lattices (Ghaffar and Fan, 2015a; Kaushik et al., 2010). Moreover, the secondary
295 peak at $2\theta=18^\circ$ was also observed, which confirms the presence of cellulose with the form of cellulose
296 I crystal. The results also confirm that the peak intensities were slightly increased due to the pre-
297 treatments. **Table 1** illustrates the crystallinity index of untreated and pre-treated straw samples. The
298 highest crystallinity index was obtained for hot water followed by microwave (H+M) pre-treatment,
299 i.e. **52%**, compared to the untreated straws, i.e. **48%**. The crystallinity index increase generated by the
300 pre-treatments can be attributed to the extraction of fundamentally amorphous polymers within the
301 constituents of wheat straw, i.e. lignin and hemicellulose (Alemdar and Sain, 2008a). Similar results
302 have been reported on bamboo fibres by Fatriasari et al. (2016). Their results indicated that the
303 microwave power deteriorate the lignin network along with the removal of hemicellulose within the

304 substrate structure by molecular vibration. Moreover, the microwave irradiation increases the total
 305 crystalline cellulose contents of the surface of the natural fibres, hence increases the crystallinity index
 306 percentage (Fatriasari et al., 2016).

307 **Figure 8**

308 **Table 1** – Crystallinity index of wheat straw internode samples with and without pre-treatments.

Sample	Crystallinity index (%)	Crystallinity index change (%)
UN	48(±0.3)*	N/A
H + S	50(±0.1)	5.0
0.5H + S	50(±0.5)	4.0
H + M	52(±0.1)	9.0

*Standard deviations are given in parentheses.

309 **3.4 Tensile properties improvements of wheat straw bio-based composites**

310 The tensile strength properties assessments, including elastic modulus and tensile toughness, and
 311 **elongation at break** have been carried out to investigate the efficacy of pre-treatments on the
 312 mechanical proficiency of bio-based composites. The coefficient of variation (CoV %) for all the
 313 compositions was calculated, which varied from 1% to 7.5%, to define the statistical significance.

314 Prior to tensile testing, the average density of bio-based composites was calculated, as shown
 315 in **Figure 9**. The results revealed that the presence of pre-treated straw in bio-based composites
 316 induces a negligible reduction in density compared to that of bio-based composites with untreated
 317 straw. The term high-density fibreboard composites refers to the composites with the density higher
 318 than 0.8 g/cm³(Guo et al., 2020; Mamiński et al., 2018). In this study, the density values of all the bio-
 319 based composites have been varied between 0.92 g/cm³ and 0.96 g/cm³, which indicates the high-
 320 density characteristic of the manufactured bio-based composites.

321 All the bio-based composites with pre-treated straws showed an increase in strength with respect to
 322 the untreated sample (see **Figure 9a**). The tensile strength increased from 3.75MPa for untreated
 323 wheat straw bio-based composites samples to 10 MPa, 6.6 MPa, and 7.9 MPa for H+S, 0.5H+S, and
 324 H+M bio-based composites samples, respectively. **Furthermore, elastic modulus, tensile toughness,
 325 and elongation at break increased for bio-based composites with the pre-treated straws, see Figure
 326 9b. The maximum enhancement compared to untreated straw bio-based composites was observed in
 327 H+S samples with 68%, 285%, and 18% elastic modulus, tensile toughness, and elongation at break
 328 respectively.** The remarkable increase in the mechanical performance of pre-treated over untreated
 329 bio-based composites could be attributed to several factors. First, the wetting property improvements

330 leading to better interfacial bonding between polymer matrices and straw (Rakesh Kumar¹, Sangeeta
331 Obrai¹, 2011). Second, the cell walls expansion during the steaming stage (S), which allows more
332 polymer to penetrate inside the swollen straw cells and hence increase the bonding. Third, increased
333 crystallinity index, as observed in the X-ray diffraction spectroscopy, indicating the strength
334 improvement of pre-treated samples due to the presence of more stable cellulose chains in their
335 structure (Xiao et al., 2014). Apart from better surface functionality and interaction with PLA, the
336 author's previous studies showed that the tensile strength of H+S pre-treated individual strands
337 significantly increased by 35%, reaching 88.9 MPa compared to the untreated strands (i.e. 66 MPa)
338 (Ghaffar et al., 2017a). All of the points mentioned above explain the improved tensile performance
339 of pre-treated straw bio-based composites compared to the untreated counterparts.

340 **Figure 9**

341 **3.5 Dimensional stability of pre-treated and untreated wheat straw bio-based composites**

342 The dimensional stability of manufactured bio-based composites was assessed by employing water
343 absorption (WA), and thickness swelling (TS) tests at 2 and 24 hours. Both WA and TS of all bio-based
344 composites, with and without pre-treated straws increased by time (see **Figure 10 a-b**). The WA results
345 indicated that, at 2 hours of immersion, the water intake was increased from 10% (CoV 3.6%) for
346 untreated straw bio-based composites to 14% (CoV 2.1%), 14% (CoV 5.2%) and 13% (CoV 4.2%) for
347 H+S, 0.5H+S, and H+M, respectively. A comparable trend has been observed for the samples after 24
348 hours of immersion in water (**Figure 10 a**). The same effect was observed for TS of bio-based
349 composites, increasing from 3% (CoV 6.5%) and 7% (CoV 6.5%) for untreated straw bio-based
350 composites to 6% (CoV 3.5%) and 8% (CoV 2.5%) for H+S counterparts at 2h and 24h, respectively
351 (**Figure 10 b**). The bio-based composites with H+S pre-treated straws exhibited maximum water
352 absorption at both 2h and 24h. This could be attributed to a decrease in the hydrophobicity of straw
353 strands after pre-treatments, due to diminished levels of wax and silica content (Kalagar, M., Bazayar,
354 B., Khademieslam, H., Ghasmi, and Hemmasi, 2015). Generally, water absorption of commercial
355 boards is approximately 51% (Jabeen et al., 2014), and maximum thickness swelling after 24 hours
356 immersion time is 15%, according to BS EN 312:2010 standard. Moreover, as reported by Zheng et al.
357 (2018), the pre-treatments increased the porosity of straw biomass surface structure by partial
358 degradation of hemicellulose and lignin (Wu et al., 2017; Zheng et al., 2018), which allowed the water
359 molecules to penetrate the bio-based composites. This resulted in high water absorption, which
360 subsequently led to increased thickness swelling. The results are aligned with the results reported by
361 Melo et al. (2014), which showed remarkable loss of dimensional stability in rice husk and bamboo
362 particles by 60% in terms of both WA and TC after 2h and 24h of immersion (Melo et al., 2014). It has

363 been reported that the chemical pre-treatments reduce water absorption (Mishra and Sain, 2009) as
364 a result of the reaction of acetyl groups with hydroxyl groups (Sahai and Pardeshi, 2019).

365 **Figure 10**

366 **4. Conclusions**

367 The main objective of this study was to develop a wheat straw bio-based composites, using optimised
368 wheat straw particles as reinforcement and PLA as polymer matrix. Several characterisation methods
369 have been carried out, and eventually, mechanical and physical properties of manufactured bio-based
370 composites reinforced with pre-treated and untreated wheat straw were assessed. The main
371 concluding remarks have been drawn out in the following points:

- 372 (i) Microstructure changes were observed in all the pre-treated wheat straw, where the pre-
373 treatments led to an increased size of parenchyma. This induced change could then
374 potentially result in more intimate polymer matrices and wheat straw entanglement,
375 therefore higher tensile strength of bio-based composites is observed.
- 376 (ii) The surface chemical functional group analysis revealed that pre-treatments reduced the
377 silica and wax from the outer surface of wheat straw, which reduced hydrophobicity of
378 the surface. Moreover, H + S and 0.5H + S pre-treatments reduced hemicellulose levels,
379 which adversely contributes to the water permeability of the bio-based composites.
- 380 (iii) The crystallinity index evaluations confirmed the changes of hemicellulose and lignin
381 levels. It was observed that H + M pre-treatment has the highest increase in crystallinity
382 index ,i.e. 9%, of wheat straw compared to untreated samples.
- 383 (iv) H + S pre-treatment drastically increased tensile strength, elastic modulus and toughness
384 of manufactured bio-based composites by 166%, 68 %, and 285%, respectively compared
385 to untreated wheat straw bio-based composites.

386 It can be concluded that H+S pre-treatment exhibited the maximum enhancement in mechanical
387 properties of wheat straw polylactic acid (PLA) bio-based composites. However, the enhanced surface
388 properties of wheat straw generated by the aforementioned pre-treatment drastically decreased the
389 dimensional stability of final products, there are possible strategies to overcome this shortcoming
390 which will be reported in future study.

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